

Effect of Rare Earth Upon As-Cast Microstructure and High Temperature Performance of ZA40 Alloy

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Abstract: In testing comparison, this study is made on the modification effect of 0.05 wt.% La-rich rare earth (RE) upon as-cast microstructure, mechanical properties of materials at high temperatures (100, 150, 200°C) and wear resistance under different temperatures (100 and 200°C) of ZA40 alloy containing Si. The addition of RE shows that it can not only effectively refine Al-rich and Cu-rich phase, but also reduce the amount of needle-like eutectic silicon, and then make the size of massive primary silicon phase smaller. Thus, there are the improvement of strength, ductility and wear resistance of materials at high temperature. The analyses are also conducted on the mechanism of underlying RE elements.

Keywords: High temperature performance, high temperature wear resistance, microstructure, mixed rare earth, ZA40.

1. INTRODUCTION

High aluminum zinc-based alloys are increasingly popular because they are characterized by higher strength and hardness, the improved wear resistance and high intrinsic damping capacity, lower coefficient and temperature difference of friction, and lower manufacturing cost [1-3]. However, there are the demerits of the low melting point and eutectic temperature of zinc and the wide range of solidification, with actual solid-phase point of about 380°C. It causes the alloy at high-temperature to have low strength. Moreover, the solubility of Al-Zn-based α phase varies greatly with an increase of temperature. Without stable organization and with the limitations of the improvement to high-temperature performance, thus greatly reducing the use of high aluminum Zn-based alloy. It is important to improve the high-temperature properties of Zn-based alloys with the spreading applications of it [4, 5].

Rare earth (RE) elements are typical metals with the active chemical properties, allowing interaction with all the elements except inert gas. The strengthening effect of RE have widely been applied to the ferrous materials and the mechanism underlying this process has been studied. However, a few researches have been conducted for the effect of RE upon non-ferrous metal [6-9].

With comparative methods, the effects of RE have been investigated upon microstructure, mechanical properties and wear resistance of ZA40 alloy containing Si with 0.05 wt.% rich-La mixed RE. The research can improve high-

temperature performances of the alloy, thereby broadening its application.

2. EXPERIMENTAL PROCEDURES

2.1. Materials

The material compositions of the ZA40 alloy with 0.05 wt.% RE (material A) and the ZA40 without RE (material B) are listed in Table 1.

The alloy was melted in a resistance furnace of graphite crucible (2.5 kg alloy for each furnace). The intermediate alloy melting method was utilized to decrease the melting temperature, oxidation of elements, and burning loss, while RE elements in the form of Al-10% RE compounds were added. Metal (cast iron) casting method was adopted with pouring temperature of 650°C [9]. However, oversize coarse grains and the zinc smoke (zinc-oxide) occurred when the liquid temperature exceeded 700°C.

2.2. Mechanical Tests

Specimens for the tensile tests were in a diameter of 6 mm and in an active gauge length of 30 mm according to GB4338-84 Chinese National standard. Tensile tests were conducted in a WE-100 Hydraulic Universal Testing Machine in an environmental chamber (temperature deviation $\leq \pm 3$). Specimens were heated up to 100, 150, and 200°C and incubated for 30 min.

High-temperature wear test was conducted by using an MG-2000-type testing machine with a pin-on-disk apparatus. Test specimens (with size of $\Phi 6\text{mm} \times 12\text{mm}$) were cut from the high aluminum zinc-based alloy with and without RE. The wear plate material was 40 Cr (quenching and low-temperature tempering, 54 HRC) with a size of $\Phi 70\text{mm} \times 10\text{mm}$. The experiments were performed under the

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Table 1. Chemical composition of the materials [wt %].

Materials	Al	Cu	Mg	Si	Rare Earth	Zn
Material A	40	3-5	0	1-2	0.05	Balance
Material B	40	3-5	0	1-2	0	Balance

stationary normal loads of 150 N, and sliding speed of 150 rpm at 100 and 150°C, respectively and both of them were in the states of dry friction. All the specimens were taken for the running test for 10min and then steady wear-in test for 30min. The amounts of weight loss before and after the tests were weighed by using electronic balance E180 (with accuracy of ± 0.1 mg) in terms of weight differences of delta m.

During the tests, JSM-7001F scanning electron microscope and Nikon-300 microscope were utilized to examine the worn surface and microstructures, so as to verify the dominant wear mechanisms.

3. RESULTS AND DISCUSSION

3.1. Effects of RE on Microstructure of Material

As-cast microstructures of materials with RE and without RE are shown in Fig. (1).

As shown in Fig. (1), the addition of RE refined the microstructures of the materials. Aluminum-based α dendrites of material B with RE (40 μm) were uniformly distributed in the matrix, while the particle distribution (size of approximately 20 μm to 30 μm) of ZA40 alloy without RE was disordered. Due to the addition of mixed RE to ZA40 alloy; the disordered particles have formed a large number of dispersed heterogeneous cores (such as high melting point CeAl4 phase). Moreover, due to the mutual constraint of grain growth, the crystallization of the heterogeneous cores was resulting in its' difficult growth into coarse dendrites. There another merit to enrich RE element at the forefront of solid-liquid interface during solidification, thus affecting the solute redistribution and impeding the atom diffusion of the alloy. Therefore, it made more and more undercooling and

the growth of retarding α -phase were impeded, while the wall spaces of secondary dendrite were reduced. Meanwhile, the uniform distribution of Cu-rich phase appeared both with an increase of grain boundaries and reduction of wall spaces in the secondary dendrite.

It can be found from Fig. (1) that the morphology of Si phase has significantly varied. The Material B (without RE) comprised a large number of acicular eutectic silicon phases and massive primary Si phases, while the precipitation and amount of eutectic silicon phases were restrained and decreased in the number with the addition of RE on Si phases (especially on eutectic Si phase). Metamorphism of the primary Si phases was also refined.

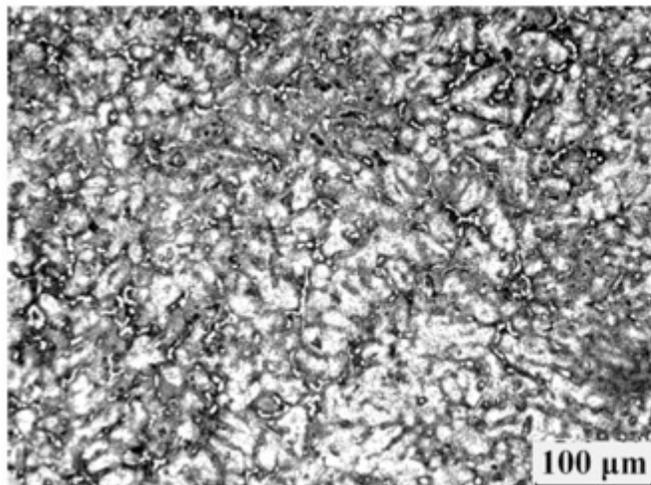
3.2. The Effect of Rare Earth Upon High-Temperature Mechanical Properties of Material

Table 2 is the strength and elongation of material A with RE, and material B without RE at different temperatures.

It can be observed from Table 2 that the addition of RE not only improved the strength and ductility of ZA40 containing Si at room temperature, but also enhanced high-temperature strength and ductility. At higher temperatures, the function of RE became more apparent.

The Strengths at ambient and elevated temperatures have been enhanced with the addition of RE. The grains at ϵ phase and Si phase were refined since the addition of RE. With the combined action of the fine grain strengthening, dispersion strengthening and fine ϵ phase effectively improves the tensile strength of the material. Hard Si phases (hardness of 1100HV) hindered dislocation movement during the tensile process. Similarly, the presence of Si phase increases the material hardness, especially at high temperatures, and

(a) Material A with RE



(b) Material B without with out RE

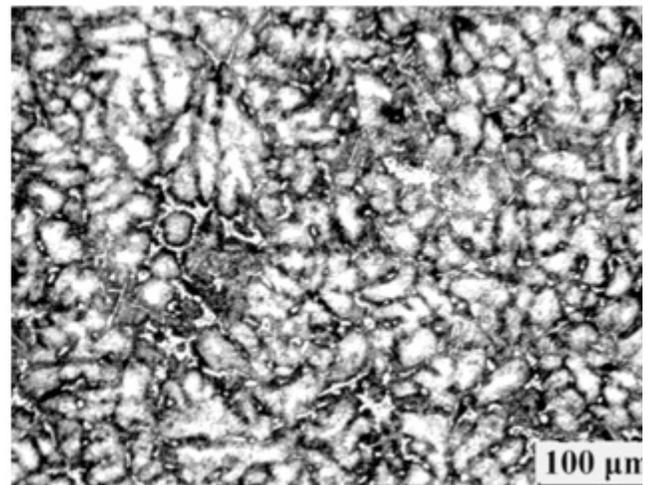
**Fig. (1).** As-cast microstructure of different zinc alloys.

Table 2. Mechanical properties of materials at different temperatures.

Material	Temperature [°C]	Yield Strength [Rm/Mpa]	Elongation A [%]
Material A	Room temperature	387	2.82
	100	354.19	14.89
	150	328.48	27.72
	200	256.05	40.92
Material B	Room temperature	377	2.48
	100	331.17	12.31
	150	299.57	22.01
	200	209.40	36.65

elevates the operating temperature of the material [10].

RE compounds can improve the high temperature properties of ZA40. The matrix of ZA40 alloy mainly consists of aluminum-based solid solution and zinc-rich phase, where the solubility of RE in Al or Zn matrix was very limited [11]. The majority of Ce and La were associated with Cu, Zn, Fe, and other elements, and they formed the complex compounds that exhibit the higher hardness and hot hardness even at high temperatures. During crystallization, the compounds are pushed to where dendrites and grain boundaries exist, where they hinder softening and grain boundary movement. Moreover, the melting points of RE elements are higher than those of Zn, Al and Cu, and much lower in the self-diffusion coefficients. Thus, RE can hinder the atomic diffusion improve the binding forces among atoms and promote high-temperature stability of the microstructure, thereby effectively improving the high-temperature strength and creep resistance of alloy [12].

The addition of RE in the alloy results in ductility at ambient and elevated temperatures. Grain refinement not only improves the strength, but also enhances the material's plasticity. ϵ phase achieved the fine dispersion in the matrix, thus eliminating the overall deformation of materials caused by larger blocks and improved its plasticity. RE inhibited the growth of acicular eutectic Si phase precipitated in the form of primary silicon, so as to remarkably reduce the fragmentation effect on the matrix. Diversification in morphology and size of Si phase played the major role in the integration with the matrix, and also enlarged the plastic deformation of the matrix in the tensile of the materials [13].

3.3. Effects of RE Upon High-Temperature Wear Resistance of Materials

Table 3 is the wear test results of the experimental materials with RE (material A) and without RE (material B) at different temperatures.

In Table 3, the wear volume has significantly increased at 100 and 150°C, compared with that at ambient temperature.

However, RE addition improved the wear resistance at both ambient and elevated temperatures.

Under 150 N loads at 100 and 150°C, the surface morphologies of specimens after the wear test are as shown in Fig. (2).

Fig. (2) shows that the material with RE resulted in the abrasive wear because of the following factors. The first reason was the shedding of hard ϵ phase and Si phase which occurred in the contact surface between the two samples during the wearing process. Secondly, RE refined the grains of ϵ phase and Si phase to form the smaller abrasives. The high strength of material led to the shallow and smoothing of the furrowed edges without accumulation and grooves. This phenomenon accounts for the small volume of wear loss. However, the grooves in the material without RE became wider and glossier, and caused the distortion and cutting in the direction of wear. This is because the hard phase and Si phase were coarser, and the strength of the material without RE was relatively low and the plastic flow occurred to different extents in the front of the abrasives. The groove edges were crushed, accompanied by tearing phenomenon under the press. In addition, features of adhesive wear emerged in the local area. Thus, the volume of wear was large and the materials formed the patterns of abrasive and adhesive wear.

At 150°C, wear surface showed almost no furrow but exhibited the varying degrees of flaking. White and black regions of RE and non-RE samples were subjected to EDS composition analyses. The results of EDS composition analysis (Table 4) show that the wear surface may be covered by the mixed oxide because of the large amount of oxygen atoms. Between black and white areas, the differences in oxygen quantity can prove that the surface exfoliation is oxide. It can be seen from this that the oxidation wear and oxide spall wear occurred in the material surface at 150°C. In addition to oxygen atoms, C and Fe atoms were present in the worn surface. Cr atom was even observed on the worn surface of the alloy without RE,

Table 3. The results from wear test of material.

Temperature [°C]	Load [N]	Speed [rpm]	Wear Loss [g]	
			Material A	Material B
Ambient	150	150	0.0043	0.0056
100	150	150	0.0115	0.0178
150	150	150	0.0233	0.0362

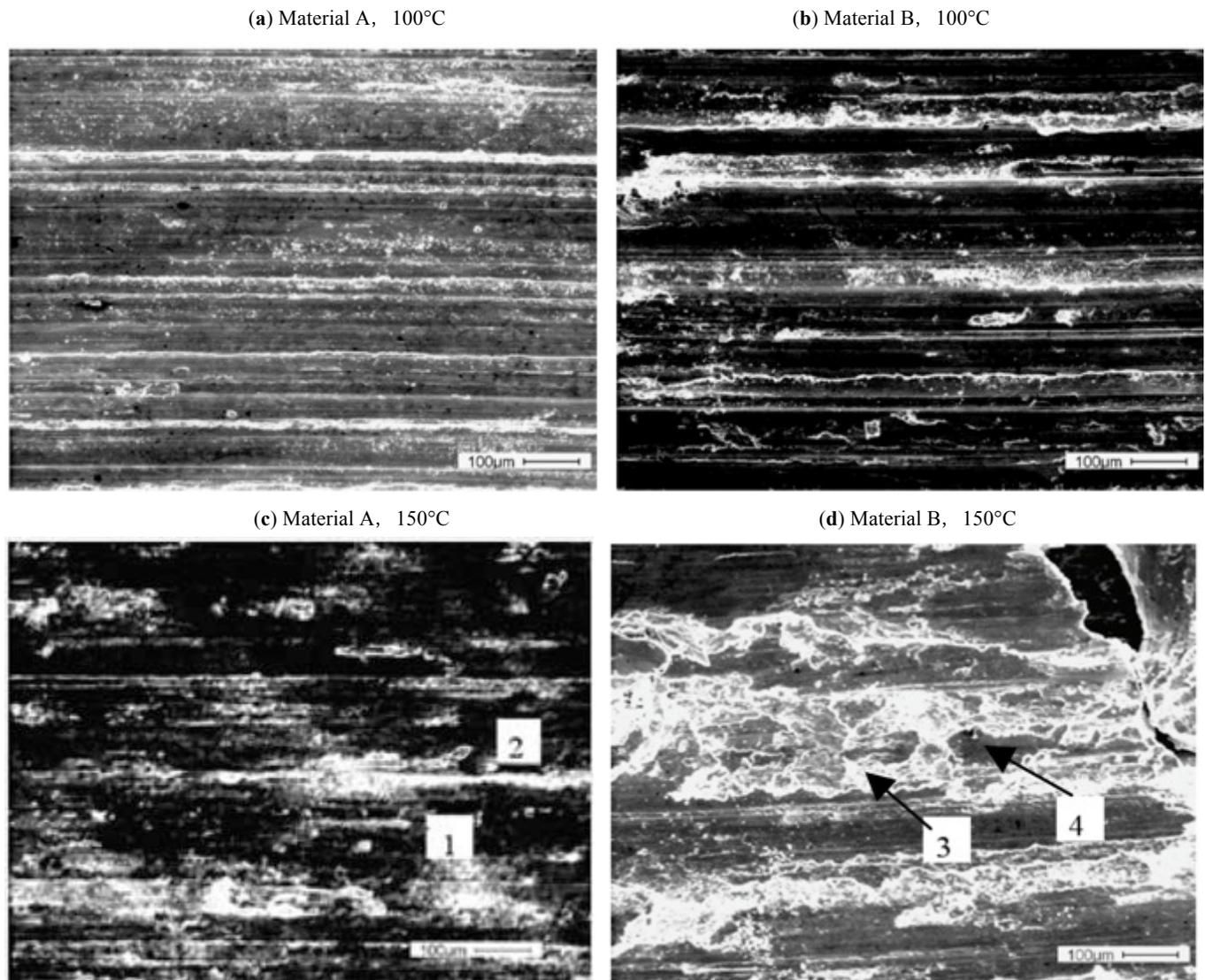


Fig. (2). Worn surface morphology at various temperatures for high aluminum zinc-based alloy with the same load and sliding speed.

obviously attributed to the wear plate material (40Gr). It is shown that C, Fe, and Gr were diverted during the wearing process. The degree of hardness of the surface oxide was too high, that it caused debris of the worn plate (40Cr) to fall on the specimen surface.

Generally, oxide films can protect the substrate and exhibit anti-friction activity. However, a matrix with the characteristics of heat resistance and thermal stability is required. The stability support of the substrate to the oxide is guaranteed during the process of wearing [14, 15].

Comparing Fig. (2c) with Fig. (2d), it can be easily found that the surface of the alloy without RE exhibited severe flaking, while comparing that with RE, which possesses only a small area of the shallow peeling.

Fig. (3) is the sectional morphology of the two wear specimens after wearing at 150°C.

It can be observed from Fig. (3) that a layer of mixed oxide is different from the matrix on the surface of the samples. Due to its low matrix strength, alloy without RE cannot support the oxide layers with high hardness and at

Table 4. Composition of worn surface.

		Atom Percentage								
		Al	Zn	Cu	Si	O	C	Fe	Cr	Total
Material A	1	37.13	14.89	1.44	2.22	34.16	1.12	9.04	0	100
	2	22.35	10.19	1.10	2.08	50.97	1.03	12.28	0	100
Material B	3	21.16	14.23	0.56	0.91	42.54	0.61	19.68	0.31	100
	4	12.06	5.54	0.41	0.73	59.11	0.49	21.07	0.59	100

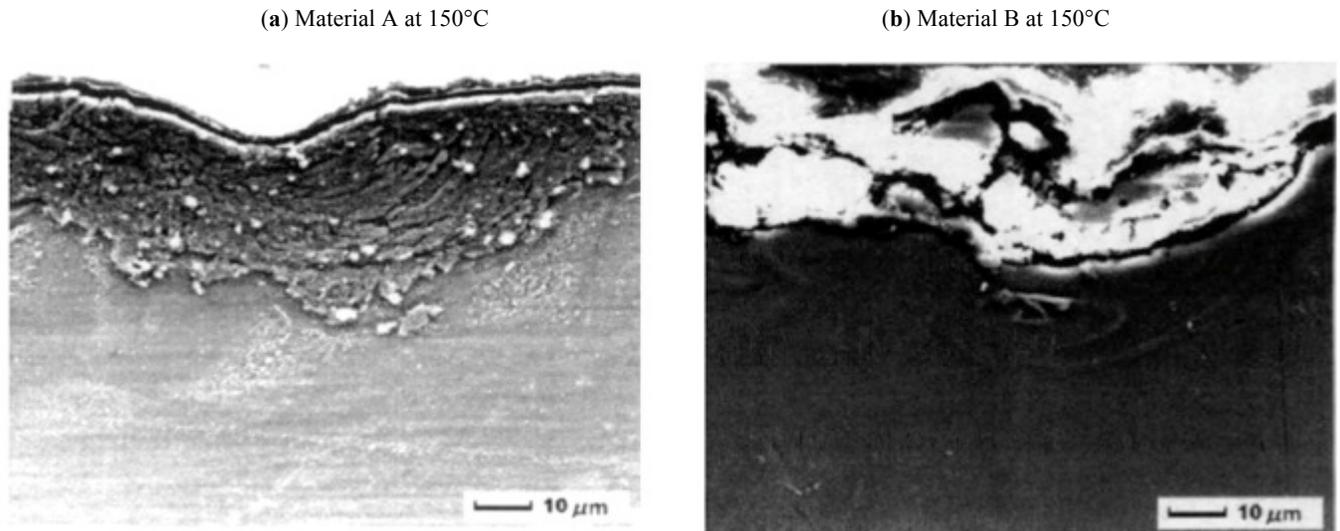


Fig. (3). Worn morphologies of ZA40 alloy samples with different amounts of RE.

high temperature. Oxide layer of samples without RE exhibited fragmentation. With the application of stress, the matrix had a larger plastic deformation, but the brittle oxide could not simultaneously arise the deformation of the matrix but cracks to be satisfied for the deformation in the matrix. Meanwhile, relative displacement caused the weak binding between substrate and oxide, even in detachment of the oxide from the substrate. Finally, large peelings were formed during cyclic loading. In the upper right corner of Fig. (2d), the matrix of the material was visible after the oxides were peeled off, showing the serious wearing. By contrast, the addition of RE in the high strength matrix of the alloy can sufficiently support the layer of oxide, thus mitigating plastic deformation under stress. The oxide layer can be satisfactory in combination with the matrix. It can be observed from Fig. (3a) that the oxide layers were tightly bonded to the substrate. Only few cracks existed within the oxide, so that only small peelings appeared during cyclic loading. In addition, the amount of wear loss was smaller than that of the material without RE.

CONCLUSION

1. The addition of 0.05 wt.% RE to ZA40 alloy containing Si not only effectively refined the aluminium-based α dendrites of as-cast microstructure, uniformly distributed the Cu-rich ϵ phase, but also retained the precipitation amount of eutectic silicon, and refined the primary silicon.
2. The addition of 0.05 wt.% mixed RE to ZA40 alloy containing Si enhanced the strength and ductility of alloy at ambient and elevated temperatures. The beneficial effects of RE were gradually apparent with an increase of temperature.
3. Under dry friction at 100°C, the alloy without RE exhibited abrasive and adhesive wear, while the addition of RE resulted in the formation of abrasive wear. Under the dry friction at 150°C, both kinds of alloys exhibited oxidative and oxide fatigue wear.
4. The addition of RE in the alloy can always improve the wear resistance of the materials, irrespective of

the temperature (100°C or 150°C). Since the addition of RE can refine the microstructure of the materials and improve the mechanical properties of materials..

CONFLICT OF INTEREST

The authors confirms that this article content has no conflict of interest.

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